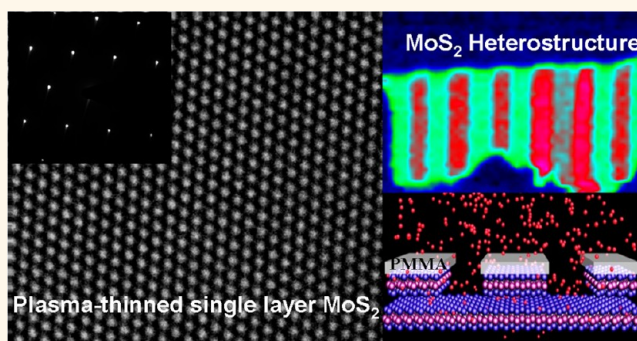


Layer-by-Layer Thinning of MoS₂ by Plasma

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ABSTRACT The electronic structures of two-dimensional materials are strongly dependent on their thicknesses; for example, there is an indirect to direct band gap transition from multilayer to single-layer MoS₂. A simple, efficient, and nondestructive way to control the thickness of MoS₂ is highly desirable for the study of thickness-dependent properties as well as for applications. Here, we present layer-by-layer thinning of MoS₂ nanosheets down to monolayer by using Ar⁺ plasma. Atomic force microscopy, high-resolution transmission electron microscopy, optical contrast, Raman, and photoluminescence spectra suggest that the top layer MoS₂ is totally removed by plasma while the bottom layer remains almost unaffected. The evolution of Raman and photoluminescence spectra of MoS₂ with thickness change is also investigated. Finally, we demonstrate that this method can be used to prepare two-dimensional heterostructures with periodical single-layer and bilayer MoS₂. The plasma thinning of MoS₂ is very reliable (with almost 100% success rate), can be easily scaled up, and is compatible with standard semiconductor process to generate heterostructures/patterns at nanometer scale, which may bring out interesting properties and new physics.



KEYWORDS: MoS₂ · plasma-induced thinning · Raman · photoluminescence · contrast

Two-dimensional layered materials have received great attention since the first experimental discovery of graphene in 2004.^{1–4} Among them, single-layer and multilayer MoS₂ have exhibited interesting properties compared to its bulk form⁵ and other two-dimensional materials. MoS₂ devices exhibit a room-temperature current on/off ratio of $\sim 1 \times 10^8$ (due to the large intrinsic band gap) and carrier mobility up to $200 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$.^{6–8} An indirect to direct band gap transition from multilayer to single-layer MoS₂ was also reported, which results in great enhancement of photoluminescence (PL).^{9,10} More interesting phenomena were presented; for example, electronic structure enables valley polarization,^{11–13} robust mechanical properties, and strain tunable band gap.^{14–16} The fabrication of large-scale MoS₂ film using chemical vapor deposition^{17–19} and a large quantity of MoS₂ nanosheets using solution-based chemical exfoliation^{20,21} also represents important

steps for its wafer-scale application in electronic devices and flexible and transparent optoelectronics.

The electronic properties of two-dimensional layered materials are strongly dependent on their thicknesses.^{4,22} Single-layer MoS₂ has a direct band gap of $\sim 1.9 \text{ eV}$, which makes it promising for application in optoelectronic devices, such as photodetectors,^{23,24} photovoltaics,¹⁶ and light emitters.²⁵ In addition, multilayer MoS₂ may also provide different and interesting properties, similar to what happened in multilayer graphene.^{22,26,27} The thickness-modulated optical energy gap of MoS₂ is desirable for phototransistors.²⁴ Electronic devices based on multilayer MoS₂ were also reported.^{7,28,29} Therefore, a simple and efficient approach to control the thickness of MoS₂ is highly desirable. It will also enable the study of thickness-dependent properties of MoS₂. Recently, Castellanos-Gomez *et al.* presented an exciting work using a

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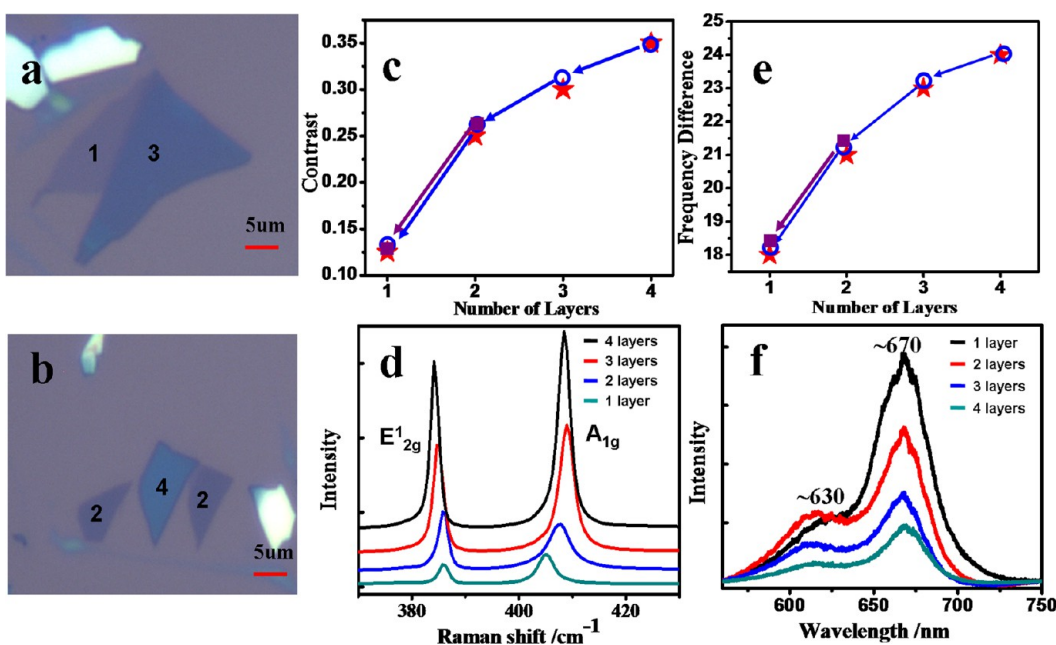


Figure 1. (a,b) Optical images of 1–4 layer MoS₂. (c) Contrast values of pristine 1–4 layer MoS₂ (red stars). Purple squares/blue circles represent the contrast values of a bilayer/quadrilayer MoS₂ sheet before and after plasma thinning. (d) Raman spectra of 1–4 layer MoS₂. (e) Frequency differences of E_{2g}¹ and A_{1g} peaks of pristine 1–4 layer MoS₂ (red stars). Purple squares/blue circles represent the frequency differences of a bilayer/quadrilayer MoS₂ sheet before and after plasma thinning. (f) PL spectra of 1–4 layer MoS₂.

laser to thin MoS₂ sheets down to monolayer thickness.³⁰ However, the origin of laser thinning is thermal ablation which makes it only possible to produce single-layer MoS₂ and challenging for scale-up.^{4,30} It would also be difficult to use this method to make high-resolution patterns (e.g., down to nanometer size), due to the optical diffraction limit of the laser spot.

Here, we present layer-by-layer thinning of multi-layer MoS₂ down to the monolayer by using Ar⁺ plasma. The thickness change is confirmed by atomic force microscopy (AFM), transmission electron microscopy (TEM), optical contrast, Raman, and PL spectra. The top layer MoS₂ is totally removed by plasma, while the bottom layer remains almost unaffected. We also demonstrate that, combined with standard lithographic techniques, this method can be used to prepare two-dimensional MoS₂ heterostructures with different thicknesses and sizes. A heterostructure composed of periodical single-layer and bilayer MoS₂ is presented. This method enables large-scale thickness control of MoS₂ and other two-dimensional materials.

RESULTS AND DISCUSSION

Figure 1a,b shows the optical images of MoS₂ sheets with a thickness of 1–4 layers. The optical contrast of MoS₂ increases with the increase of its thickness. In our previous work, we proposed to use the contrast between R/G/B values of the sample and substrate in an optical image to precisely determine the thickness of two-dimensional materials.³¹ Here, the R channel

contrast is used to determine the thickness of MoS₂ with the following equation:

$$C = \frac{R_{\text{sub}} - R_{\text{sam}}}{R_{\text{sub}}} \quad (1)$$

where C is the contrast of the MoS₂ sheet, R_{sub} and R_{sam} are the R channel values of the SiO₂/Si substrate and MoS₂ sheets obtained from the optical image. The contrast values for 1–4 layers of MoS₂ are 0.125, 0.25, 0.31, and 0.36 (red stars in Figure 1c), which agree quite well with the results obtained by Li *et al.* (R value differences between sample and substrate are 28, 56, and 72 for 1, 2, and 3 layers of MoS₂, respectively).³² In our work, the use of contrast to determine the layer thickness is more quantitative as it does not depend on the intensity of incident light and parameters of the microscope. However, care should be taken when using contrast to determine the thickness of two-dimensional materials. The contrast value could be affected by the thickness of an oxide capping layer; for example, the contrast values of single-layer MoS₂ on 300 and 285 nm SiO₂/Si substrates are ~ 0.125 and ~ 0.11 , respectively. The thickness of MoS₂ is further confirmed by Raman spectra. Raman spectroscopy has been widely used to study two-dimensional materials and to identify their thicknesses.^{33–36} The Raman spectra of 1–4 layer MoS₂ are shown in Figure 1d. Two main Raman features are clearly presented, which correspond to E_{2g}¹ (~ 385 cm⁻¹, in-plane vibration of two S atoms with respect to the Mo atom) and A_{1g} (~ 405 cm⁻¹, out-of-plane vibration of S atoms)

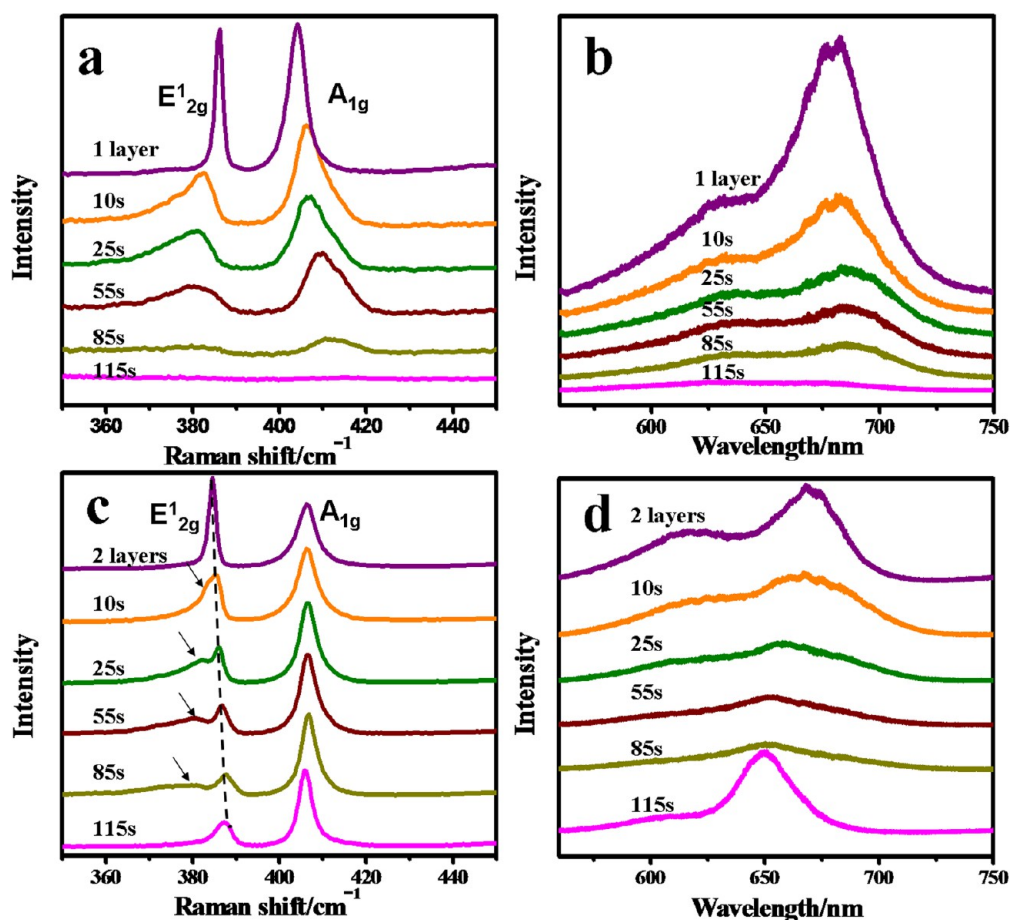


Figure 2. Raman (a) and PL (b) spectra of single-layer MoS₂ after Ar⁺ plasma irradiation with different time. Raman (c) and PL (d) spectra of bilayer MoS₂ after plasma irradiation with different time.

modes.³⁵ The intensities of these two peaks increase with the increase of thickness. In addition, the frequency of the E_{2g}¹ peak decreases and that of the A_{1g} peak increases with the increase of thickness, due to the change of force constant and also variation in the dielectric screening environment.^{35,37} The frequency difference of these two peaks can then be used to identify the thickness of MoS₂ sheets. The red stars in Figure 1e show the frequency differences of E_{2g}¹ and A_{1g} peaks, and the values are 18, 21, 23, and 24 cm⁻¹ for 1–4 layer MoS₂, respectively. These results agree quite well with previous reports.^{34–36} Figure 1f shows the PL spectra of 1–4 layer MoS₂. The PL spectra contain two peaks located at ~670 and ~630 nm, corresponding to the direct excitonic transitions between the minimum of the conduction band and the maxima of split valence bands (A1 and B1 excitons) in MoS₂.^{9,10} The single-layer MoS₂ has a direct band gap structure; therefore, it has the strongest PL intensity. The PL intensity decreases with the increase of thickness.¹⁰ The intensity difference between single-layer and multilayer MoS₂ is not as distinct as that reported by Mak *et al.*⁹ but is similar to the result of Splendiani *et al.*¹⁰ The PL intensity of single-layer MoS₂ can be greatly enhanced (by more than 10 times) after

annealing in 300 °C in vacuum, while that of multilayer MoS₂ does not show noticeable changes. This might be the reason why different relative PL intensities of single-layer and multilayer MoS₂ were observed by different groups. The contrast values, Raman frequency differences, as well as PL intensities were used to monitor the thickness change of MoS₂ sheets after Ar⁺ plasma irradiation.

Figure 2a,b shows Raman and PL spectra of single-layer MoS₂ after Ar⁺ plasma irradiation (0–115 s). The Raman peaks become weak and broadened after plasma irradiation (10–85 s), which suggests that MoS₂ becomes disordered. After 115 s irradiation, the Raman peaks disappear, indicating that the single-layer MoS₂ is totally removed by plasma. The PL intensities of MoS₂ become much weaker with the increase of irradiation time (10–85 s), and it almost disappears after 115 s irradiation. The changes of Raman and PL spectra suggest that single-layer MoS₂ becomes disordered and can finally be removed by Ar⁺ plasma. For bilayer MoS₂, the results are quite different. As can be seen in Figure 2c, the E_{2g}¹ peak of bilayer MoS₂ splits after 10 s irradiation, as denoted by the black arrows. The emerging peak at ~380 cm⁻¹ becomes broader and weaker (10–85 s) and finally

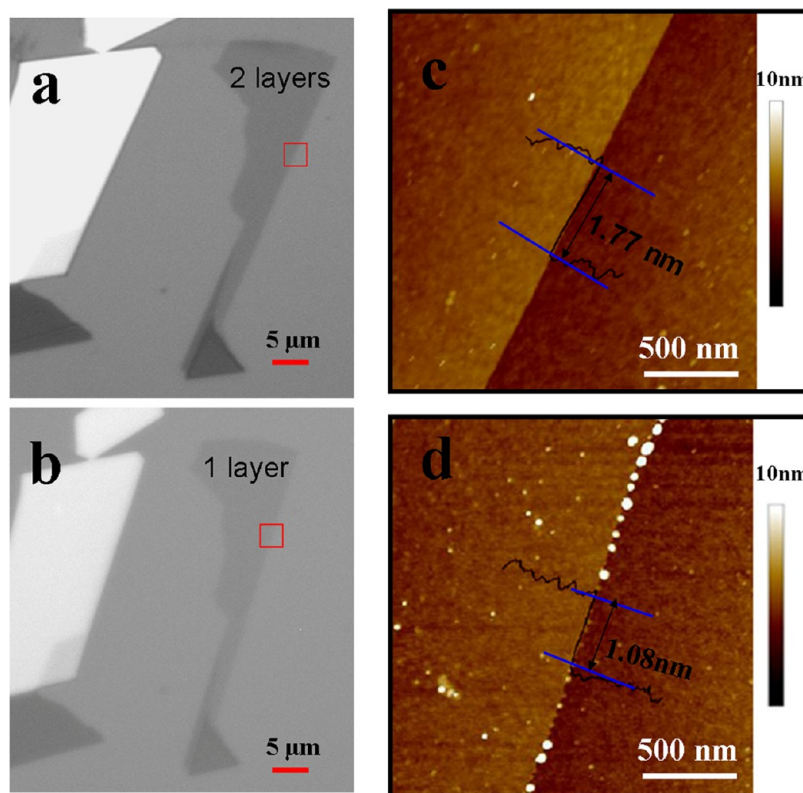


Figure 3. *R* channel optical images of a bilayer MoS₂ before (a) and after (b) Ar⁺ plasma thinning. AFM images of the same sample before (c) and after (d) plasma thinning.

disappears (115 s) with the increase of irradiation time, similar to the E_{2g}¹ peak of plasma-irradiated single-layer MoS₂ (Figure 2a). This suggests that the emerging peak belongs to the top layer MoS₂. Another interesting phenomenon is that, with the increase of irradiation time, the E_{2g}¹ peak (the sharp peak belongs to the bottom layer) of bilayer MoS₂ shifts to higher frequency, toward the E_{2g}¹ frequency of pristine single-layer MoS₂ (~385 cm⁻¹). It has been theoretically explained that the decrease of E_{2g}¹ frequency with the increase of MoS₂ thickness is due to the stronger dielectric screening of the long-range Coulomb interaction in multilayer and bulk MoS₂.³⁷ The Ar⁺ plasma irradiation introduces defects in the top layer MoS₂ and makes it disordered. As a result, the interlayer interaction between the top and bottom MoS₂ layers as well as the dielectric screening effect become much weaker, which results in the blue shift of E_{2g}¹ peak after plasma irradiation. After 115 s irradiation, the E_{2g}¹ and A_{1g}¹ peaks of “bilayer” MoS₂ are very sharp and the frequency difference of these two peaks decreases from 21.5 cm⁻¹ of bilayer MoS₂ to 18.5 cm⁻¹, which is the value of single-layer MoS₂ (purple squares in Figure 1e). In addition to Raman spectra, PL spectra of MoS₂ should give much clearer evidence of the thickness change, as the PL intensities of MoS₂ increase with the decrease of thickness.⁹ As can be seen in Figure 2d, the PL intensities of bilayer MoS₂ decrease with the increase of irradiation time (10–85 s), as the top layer

MoS₂ becomes disordered after plasma irradiation. After 115 s irradiation, the PL intensity has an abrupt increase, which is in agreement with the thickness change from bilayer to single-layer (indirect to direct band gap transition).

Figure 3a,b shows the *R* channel optical image of a bilayer MoS₂ before and after plasma thinning. The contrast values derived from these two images are 0.26 and 0.13 (purple squares in Figure 1c), respectively, close to the values of pristine bilayer and single-layer MoS₂. The AFM images are also presented in Figure 3c, d to study the morphologies of MoS₂ before and after plasma thinning. The thickness of pristine bilayer MoS₂ is ~1.77 nm, which is a bit larger than the theoretical value. This is a common observation for AFM images of layered materials, as there might be water or molecules trapped between the sample and substrate, and the trapping responses of the sample and substrate are also different.³⁶ The thickness of MoS₂ after plasma thinning is ~1.08 nm, with a thickness difference of ~0.69 nm compared to the pristine sample. This is quite close to the 0.62 nm interlayer spacing of bulk MoS₂.³⁸ It is worth noting that the surface of single-layer MoS₂ obtained by plasma thinning is still very smooth, with a roughness of 0.19 nm (0.18 and 0.13 nm for pristine single-layer and bilayer MoS₂), indicating that there are very few unrecovered MoS₂ traces on the surface.³⁰

High-resolution TEM is used to study the atomic structures of a bilayer MoS₂ before (Figure 4a–c) and

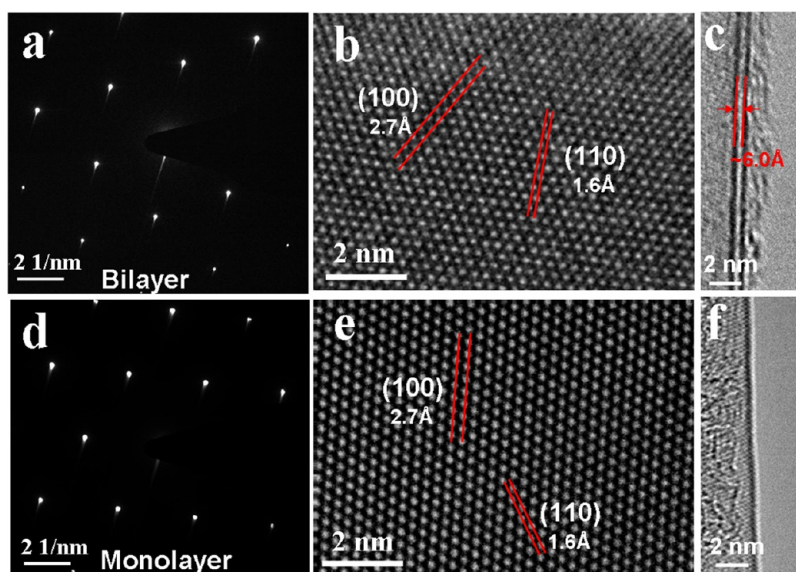


Figure 4. Diffraction patterns, high-resolution TEM images and folded edges of a pristine bilayer MoS₂ (a–c) and a plasma-thinned single-layer MoS₂ (d–f).

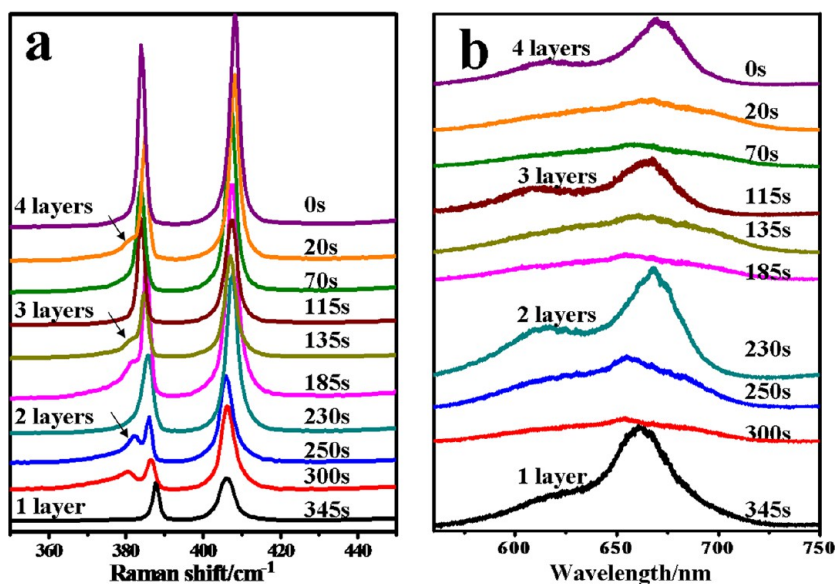


Figure 5. Raman (a) and PL (b) spectra of a the quadrilayer MoS₂ sheet after Ar⁺ plasma irradiation with different time.

after (Figure 4d–f) plasma thinning. The electron diffraction patterns (Figure 4a,d) and high-resolution TEM images (Figure 4b,e) reveal the hexagonal lattice structure of MoS₂ with a lattice spacing of ~ 2.7 Å ((100) plane) and ~ 1.6 Å ((110) plane). No noticeable difference in the lattice parameters of MoS₂ is found before and after plasma thinning within the resolution of the system. The folded edges of two-dimensional materials are frequently adopted to identify the number of layers in TEM.^{18,26,33,39} Bilayer MoS₂ presents two dark fringes with a space of ~ 6.0 Å, in good agreement with the interlayer distance of MoS₂. On the other hand, plasma-thinned single-layer MoS₂ has only one dark fringe. The most important observation of high-resolution TEM is that the structure of plasma-thinned single-layer MoS₂

remains extremely high quality (Figure 4e), indicating that the top layer MoS₂ is totally removed while the bottom layer remains almost unaffected by plasma. This may relate to the strong bonding strength of the S–Mo–S structure and the weak interlayer interaction. Plasma irradiation was also used to thin down graphene sheets.^{40–42} However, reactive sources such as H₂O₂⁴² and hydrogen⁴⁰ were used, which induced excessive defects in graphene. High-temperature (900 °C) post-annealing in O₂ was also used to etch the disordered top layer.⁴¹ The S–Mo–S bonding of MoS₂ is not as strong as that of C–C bonding of graphene,^{14,15} which makes MoS₂ more easily removed by Ar⁺ plasma. We would emphasize that the plasma thinning of MoS₂ is highly reproducible and has

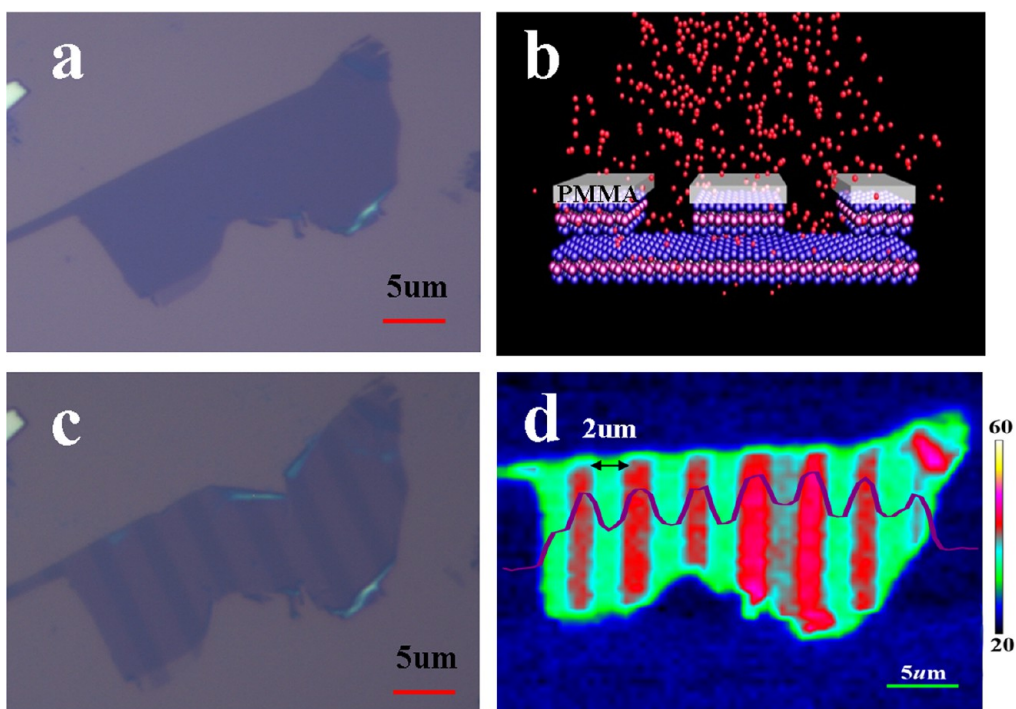


Figure 6. (a) Optical images of a pristine bilayer MoS₂. (b) Schematic showing the process of making the heterostructure. (c) Optical image of a periodical MoS₂ heterostructure obtained by plasma thinning. (d) Raman image generated by the intensities of the E_{2g} peak of the heterostructure. Inset is the cross section of the peak intensity across the sample.

almost 100% success rate for more than 10 processed samples. An important sign of thickness change is the abrupt increase of PL intensity after irradiation.

We can also do layer-by-layer thinning of MoS₂, as shown in Figure 5. A quadrilayer MoS₂ is irradiated by Ar⁺ plasma with similar conditions (0–345 s). The E_{2g}¹ peak first splits (as indicated by the black arrows in Figure 5a), and then the emerging peak disappears after 115 s irradiation. At the same time, the PL intensity has an abrupt increase (Figure 5b), indicating the removal of the topmost layer of MoS₂. Similar changes of Raman and PL spectra repeat for the removal of additional layers. The E_{2g}¹ peak blue shifts, while the A_{1g} peak red shifts with the decrease of MoS₂ thickness. The frequency difference of E_{2g}¹ and A_{1g} peaks changes from 24 cm⁻¹ (quadrilayer) to 23.4 cm⁻¹ (trilayers) to 21.5 cm⁻¹ (bilayer) and, finally, to 18.2 cm⁻¹ (single-layer) (as shown by the blue circles in Figure 1e). The *R* contrast value of MoS₂ decreases from 0.36 (quadrilayers) to 0.32 (trilayers) to 0.26 (bilayer) and, finally, to 0.125 (single-layer) (as shown by the blue circles in Figure 1c). These results suggest that the plasma thinning of MoS₂ is very reliable and can be used for devices which require structures with different thicknesses, such as phototransistors with thickness-modulated optical energy gaps. It can also be used to study the thickness-dependent properties of MoS₂ sheets. Moreover, this method can be easily scaled up for the wafer-size process. Combined with standard lithographic technique, it can also be used to fabricate continuous two-dimensional heterostructures.

As a demonstration, we fabricate a periodical 1/2/1/2/1...MoS₂ heterostructure. A pristine bilayer MoS₂ (Figure 6a) is spin-coated by a thin layer of polymethyl methacrylate (PMMA), followed by electron beam lithography to create a mask of periodical lines with a width of ~2 μm (Figure 6b). Ar⁺ plasma is then used to thin down the uncovered area to form periodical heterostructures. The optical image of the obtained structure in Figure 6c shows MoS₂ patterns with different thicknesses/contrast. The Raman image created by the intensity of the E_{2g}¹ peak is shown in Figure 6d. The red color represents higher Raman intensity (bilayer), and green color represents lower intensity (single-layer). Figure 6 demonstrates that, by using plasma irradiation and standard lithography, we can achieve arbitrary MoS₂ heterostructures with different thickness. The width of the structure can be at nanometer scale with careful control of experimental conditions. The periodical heterostructures containing materials with different electronic properties⁴³ (e.g., direct and indirect band gap structures for single-layer and bilayer MoS₂) may bring out interesting properties and new physics.

CONCLUSION

In summary, we have demonstrated that the thickness of MoS₂ sheets can be controlled layer-by-layer under Ar⁺ plasma irradiation. AFM, high-resolution TEM, Raman, and PL spectra suggest that the top layer MoS₂ is totally removed, while the bottom layer MoS₂ remains almost unaffected, which makes this method

promising for fabricating electronic devices with different structures and thicknesses. We have also prepared a two-dimensional heterostructure containing periodical

single-layer and bilayer MoS₂. The plasma thinning provides an easy and efficient way to control the properties of MoS₂ and other two-dimensional materials.

EXPERIMENTAL METHODS

Single-layer and multilayer MoS₂ are fabricated by mechanical exfoliation from single-crystal MoS₂ and deposited onto a 300 nm SiO₂/Si substrate. Optical microscopy, Raman, and AFM are used to identify the thickness. Ar⁺ plasma (commercial 13.56 MHz RF source) with power of 50 W and pressure of 40 Pa is used to thin down the MoS₂ sheets with different irradiation time at room temperature. The Raman and PL spectra are recorded using a LabRAM HR800 Raman system with 514.5 nm excitation. The laser power at the sample is lower than 0.5 mW to avoid laser-induced heating. To obtain the Raman images, an *x*–*y* stage is used to move the sample with a step size of 200 nm, and a Raman spectrum is recorded at every point. AFM is carried out with a Nanoscope 8.15 system. High-resolution TEM is carried out using an image aberration-corrected TEM system (FEI Titan 80-300). An acceleration voltage of 80 kV is chosen to achieve enough resolution while maintaining the structure of MoS₂. The MoS₂ flakes are transferred to a TEM grid using a PMMA-based transfer technique.⁴⁴ A thin layer of PMMA is spin-coated on the SiO₂/Si substrate which contains a target flake. The SiO₂ is etched away by KOH solution. The MoS₂ flake will attach on PMMA and will then be transferred to the TEM grid. An optical microscope is used to locate the flake and make aligned transfer. Finally, PMMA is dissolved in acetone, and the sample is washed by isopropyl alcohol and water several times.

Conflict of Interest: The authors declare no competing financial interest.

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